



Effects of CuO-derived catalysts morphology on carbon dioxide electrochemical reduction in a flow cell

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ABSTRACT

The electrochemical CO₂ reduction reaction (e-CO₂RR) represents a sustainable approach to convert CO₂ into valuable fuels and chemicals using renewable electricity to close the carbon loop and mitigate the climate change. In this scenario, oxide-derived Cu catalysts (*i.e.*, Cu oxides precursor modified by the application of a negative electrochemical bias to perform the CO₂ reduction) have emerged as promising materials for the e-CO₂RR, demonstrating an enhanced Faradaic Efficiency (FE) toward C₂ products, while significantly suppressing the CH₄ formation. In this work, CuO systems were synthesized using different soft-chemistry approaches (*i.e.*, precipitation vs. hydrothermal routes), and characterized through several advanced techniques such as TEM, SEM, XRD, BET analysis, Raman and FT-IR spectroscopies of adsorbed probe molecules. Catalytic performances were determined in terms of FE by monitoring the production of the main CO₂ reduction derivatives in a flow cell. By performing a systematic study, it was demonstrated that both starting CuO particles size and morphology represent critical factors determining the C–C coupling reaction, mandatory to obtain C₂ products. In particular, oxide-derived Cu catalysts presenting particles with sheet-like morphology showed superior selectivity for the CO₂ conversion into C₂ derivatives (*i.e.*, 50% at 200 mA cm^{−2}), with > 40% of C₂H₄ formation at high production rate (400 mA cm^{−2}). Interestingly, both CuO particles size increase and their morphological changes toward either tabular-prismatic or spheroidal shapes determine a substantial drop of the performances accompanied by a rise in the parasite hydrogen evolution reaction (HER).

1. Introduction

Nowadays, both the global environmental awareness [1,2] and the rapid depletion of the traditional fossil fuels [3,4] have moved the scientific community toward the search for novel, sustainable, and economically viable technologies able to exploit carbon sources

alternative to petroleum for the production of chemicals/intermediates and/or fuels, destined for the industrial chemical synthesis and the energy sectors, respectively [5–9]. Among the available carbon sources, surely the main interesting and scientifically appealing one is carbon dioxide (CO₂) [10], which is the main relevant greenhouse gas having a key (detrimental) influence on the global climate change because of the

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rapid increment of its release in the atmosphere at high concentrations, primarily due to the rapid industrialization progress [11–13].

Currently, the CO₂ molecule can be transformed into useful chemicals/fuels mainly through either catalytic [14] and photocatalytic [15] processes or biological microorganism-mediated routes [16]. Among these processes, the direct CO₂ reduction reaction through electrocatalysis (e-CO₂RR) has been extensively studied in recent years [17–20], mainly because of its high potential to transform CO₂ into many different value-added chemicals/intermediates/fuels by eventually utilizing electrons deriving from renewable energy resources (e.g., integrating this technology with photovoltaics) [21,22].

At present, many metal and metal oxide electrocatalysts (ECs) have been employed (together with supporting materials) in the e-CO₂RR with good results. Promising ECs are metallic Fe [23], Co [24], Zn [25], Ni [26], and Cu [27], and oxides such as Cu/TiO₂ [28], FeO/CoO [29], ZnO [30], NiO [31], SnO₂ [32], as well as Cu oxides [33,34].

Since the key parameters driving the e-CO₂RR processes (namely, both the ECs' selectivity, expressed as Faradaic Efficiency, FE, and the thermodynamic potential) depend on the binding strength between the EC's metal site and the adsorbed species at the surface (i.e., either CO₂ or intermediates/products), in the majority of the cases the design of the ECs is the process bottleneck. Among all the metal-based ECs, Cu is the only metal that has gained considerable attention in the field of electrocatalysis owing to its unique capacity of electrochemically reducing CO₂ into a number of oxygenates and hydrocarbon fuels, such as alcohols, aldehydes, and multi-carbon products [20,35]. In particular, Cu exhibits a proper CO adsorption capacity owing to its unique electronic structure able to promote the generation of both one-carbon (C₁) (i.e., carbon monoxide (CO) [36], methane (CH₄) [37]) and multi-carbon (C₂₊) products (i.e., ethylene (C₂H₄), ethanol (C₂H₅OH) [38,39]), together with hydrogen (H₂; HER), which is the undesired e-CO₂RR parasite reaction [40]. Therefore, there is an intrinsic selectivity issue, making the catalytic process towards the synthesis of a specific double-carbon (C₂) compound difficult to be optimized. In fact, to reach this goal, the optimization of the ECs morphology and size, by altering surface characteristics [41,42] and exposing specific crystal facets, is found to be among the potential strategies available for the enhancement of C—C coupling and C₂₊ compound production [43]. In particular, C₂₊ products (i.e., ethylene, ethanol, acetate, propanol, etc...) are very interesting chemical species from both the electrochemical and synthesis viewpoint, since they can store high-energy density and are important intermediates for a plethora of large-scale industrial processes (e.g., light olefins are essential for the manufacturing of polymeric materials), respectively [44].

Generally, nanostructured ECs show considerable advantages over their bulk equivalents, including high surface-area, improved catalytic activity, and a high density of low coordination sites for efficient e-CO₂RR [45]. The general mechanism involved in the e-CO₂RR to produce C₂₊ products is a complex and multistep process involving many surface-bound species with different reactivity [20]. Even if the exact mechanism is still unclear and likely to vary along with a variety of reaction conditions, according to the literature there are four basic steps that should be considered when modeling such heterogeneous catalytic processes, namely:

1. The chemisorption of CO₂ molecules onto the ECs surface with formation of partially charged *CO₂^{δ−} species through interactions with surface metal atoms [46]. Physisorption of CO₂, a linear molecule, may occur simultaneously, which subsequently can be activated to a chemisorbed form via surface defects, alkali-metal promoted surfaces, or the electrons generated during the electrocatalysis [47,48].
2. The electron transfer and/or proton coupling processes, breaking the C—O bond, with the formation of additional reduced intermediates.
3. The formation of a C—C bond to produce a variety of C₂₊ compounds involving the production of hydrocarbons and oxygenated forms (e.g., alcohols, aldehydes). This step is the most challenging one, as it requires a high coverage of C₁ intermediates generated during step 2.
4. The re-arrangement of the C-products followed by desorption from active sites and diffusion into the bulk electrolyte solution [49].

Operatively, in a typical electrolyzer, e-CO₂RR occurs at the cathode, while the oxygen evolution reaction (OER) takes place at the anode. The major cathodic reactions involved in the e-CO₂RR are summarized in Table 1 [20,50–53].

As previously introduced, Cu-based catalysts are very promising ECs since selective towards over a multitude of e-CO₂RR different products (namely, 16 reduction products, of which 12 being C₂₊) [54]. Furthermore, the scientific literature pointed out that the presence of a certain oxidation state of Cu significantly improves the e-CO₂RR performance towards C₂₊ product conversion, evidencing how the co-presence of surface Cu(I) and O sites provides favorable binding sites capable of modulating the ECs' reactivity and selectivity [55]. In particular, it should be noted that in e-CO₂RR, Cu catalysts in oxidized form (e.g., CuO) are “*in situ*” reduced by the application of the negative electrochemical bias into metallic Cu during the e-CO₂RR process, thus forming oxide-derived Cu catalysts (i.e., OD-Cu; Cu oxide precursors modified during the e-CO₂RR) [55,56].

Over the years, several studies have been carried out to verify the intrinsic relationship between the structure of the ECs and their catalytic properties, considering various structural features, including the EC's morphology. For instance, Ren *et al.* [57] highlighted how the presence of Cu(I) ions on the surface of the EC allows an improved selective toward the formation of C₂₊ products compared to HER, with a FE of 34.3% for C₂H₄ and 16.4% for C₂H₅OH. Theoretical calculations have also shown that, when both Cu(0) and Cu(I) sites are present on the EC's surface, the synergistic effect between these two species facilitates the CO₂ activation process and the CO dimerization, thus enhancing the process selectivity toward C₂ products [58,59]. Moreover, even the O-atoms defectivity on the surface of the ECs is a very important aspect that should be considered. In this regard, Mistry *et al.* [58] demonstrated how CuO particles treated with plasma and characterized by O-atoms defects on the surface showed improved selectivity for the CO₂-to-C₂H₄ reduction (> 60% FE).

Regarding the role played by structural and/or surface properties, ECs with different morphologies exhibit different crystal planes, which can generate a different pH near the electrode, thus consequently

Table 1

Major eCO₂RR, and relative thermodynamic equilibrium potentials with respect to the Reversible Hydrogen Electrode (RHE) in aqueous electrolyte at pH 0, 1 atmospheric pressure, and 25 °C.

Reduction products	Electrochemical reactions	Reduction potential E ⁰ (V) vs. RHE
Carbon monoxide, CO	$CO_{2(g)} + 2H^+ + 2e^- \rightarrow CO_{(g)} + H_2O_{(l)}$	−0.11
Formic acid, HCOOH	$CO_{2(g)} + 2H^+ + 2e^- \rightarrow HCOOH_{(aq)}$	−0.12
Methanol, CH ₃ OH	$CO_{2(g)} + 6H^+ + 6e^- \rightarrow CH_3OH_{(aq)} + H_2O_{(l)}$	+0.03
Methane, CH ₄	$CO_{2(g)} + 8H^+ + 8e^- \rightarrow CH_{4(g)} + 2H_2O_{(l)}$	+0.17
Acetic acid, CH ₃ COOH	$2CO_{2(g)} + 8H^+ + 8e^- \rightarrow CH_3COOH_{(aq)} + 2H_2O_{(l)}$	+0.11
Acetaldehyde, CH ₃ CHO	$2CO_{2(g)} + 10H^+ + 10e^- \rightarrow CH_3CHO_{(aq)} + 3H_2O_{(l)}$	+0.06
Ethylene, C ₂ H ₄	$2CO_{2(g)} + 12H^+ + 12e^- \rightarrow C_2H_{4(g)} + 4H_2O_{(l)}$	+0.08
Ethanol, C ₂ H ₅ OH	$2CO_{2(g)} + 12H^+ + 12e^- \rightarrow C_2H_5OH_{(aq)} + 3H_2O_{(l)}$	+0.09
Ethane, C ₂ H ₆	$2CO_{2(g)} + 14H^+ + 14e^- \rightarrow C_2H_{6(g)} + 4H_2O_{(l)}$	+0.14
Hydrogen (HER), H ₂	$2H^+ + 2e^- \rightarrow H_2$	0.00

varying both nature and concentration of reaction intermediates/products [60]. In this regard, various theoretical studies have shown that the presence of {100} crystal planes of metallic Cu is advantageous for the e-CO₂RR as favors the C—C coupling reaction leading to C₂₊ products [61]. Recently, Wu *et al.* [62] synthesized metallic Cu nanoparticles containing both Cu{100} and Cu{111} crystal planes. Electrochemical tests demonstrated an increased selectivity toward C₂₊ products (ca. 75% FE at 300 mA cm⁻² current density), compared to analogous systems containing isolated Cu{100} and Cu{111} crystal planes. Such behavior seems due to the presence of Cu{100}/Cu{111} interface, characterized by an electronic structure capable of promoting greater CO intermediate adsorption, and lowering the dimerization and C—C coupling energy. Additionally, high Miller index morphologies (e.g., ECs presenting Cu{511}, Cu{711}, and/or Cu{911} crystal planes) are characterized by a high number of under-coordinated Cu atoms, and results being more selective for C₂₊ products compared to their analogous low Miller index counterparts (e.g., Cu systems presenting only {100} crystal planes) [63,64]. Ma *et al.* [65] prepared CuO-based nanowires produced by calcining Cu(OH)₂ nanowires at 150 °C, and tested them as ECs for the e-CO₂RR, observing an increment of the selectivity toward C₂ products with the nanowire size. In general, the longer the average length of the nanowires, the lower the bicarbonate ions' diffusion in solution, and consequently the higher the selectivity toward high C-containing products. Loiudice *et al.* [66], instead, evaluated the effect of both morphology and size by comparing the electrochemical performances of metallic Cu nanospheres and nanocubes at different size, observing for both morphologies an increase in selectivity towards C₂H₄ when reducing the average particle size, with an improved selectivity toward C₂ products when employing nanocubes. Authors attributed this specific trend to the higher number of low-coordination sites available when reducing the nanocubes size. Finally, the presence of porosity and three-dimensionality are also important parameters influencing the selectivity towards C₂₊ products, as both influence the local pH, the residence time of the intermediate, the gas permeability, and the liquid diffusion rate [67].

Despite the extensive literature demonstrating the main role of Cu-based nanostructures for enhancing the e-CO₂RR, the relative contribution due to the EC's morphology, surface active sites, and gas diffusion layer organization remains poorly understood. In this regard, a plethora of scientific studies have focused on EC architectures as key descriptors governing the e-CO₂RR selectivity [17,18,20], whereas fewer studies investigated the influence of operating conditions [21,22], or structural organization, coverage uniformity, and transport properties of the EC layer on the gas diffusion electrode [28]. In this study, the present authors address this gap through a systematic investigation employing different oxide-derived Cu catalysts (*i.e.*, CuO ECs before the application of the negative electrochemical bias) prepared via scalable soft-chemistry methods, namely precipitation and hydrothermal route, with distinct morphologies, namely sheet-like, spheroidal, and tabular-prismatic structures. By combining several advanced techniques such as TEM, SEM, XRD, BET analysis, Raman spectroscopy, FT-IR spectroscopy of adsorbed CO to investigate the surface Cu active sites, and electrochemical analyses, it was evidenced the direct correlations between the initial EC morphology, exposed Cu surface sites, EC layer organization on gas diffusion electrodes, and the attained e-CO₂RR performance. Results demonstrate that the superior selectivity (quantified as FE) toward the C—C coupling reaction, mandatory to obtain C₂ products is primarily due to the 2D sheet-like morphology, whereas tabular-prismatic and/or spheroidal shapes determine a substantial drop of the performances accompanied by a rise of the undesired HER. However, such superior selectivity results from the combination of several factors, such as the surface accessibility, Cu surface environments, and the formation of a homogeneous EC network promoting C—C coupling reactions. These findings provide additional insight into the design principles governing oxide-derived Cu catalysts for selective e-CO₂RR.

2. Material and methods

2.1. Chemicals

Reactants: Copper(II) nitrate trihydrate (Cu(NO₃)₂·3H₂O, puriss. p. a., 99% purity, CAS 10031-43-3, Thermo Scientific). Solutions and solvents: different NaOH solutions were prepared by dissolving sodium hydroxide pellets (NaOH, puriss., p.a., ACS reagent, ≥ 98%, CAS 1310-73-2, Aldrich) in deionized water, whereas acetone (CAS 67-64-1, Sigma-Aldrich) was used as it is. Deionized water was used during washing procedures. Milli-Q water with a resistivity of 18.2 MΩ cm was used. All chemicals were used without further purification. Ink preparation: 2-propanol (anhydrous, 100%, CAS 67-63-0, VWR Chemicals Avantor), Nafion™ solution (PFSA 5 wt%, CAS 31175-20-9, Qintech), PET-reinforced anion exchange membrane (thickness: 130 μm, Qintech), Freudenberg H23 gas diffusion layers (Qintech). Electrochemical tests: CO₂ cylinder (5 L, 99.995%, Sapio), potassium hydroxide (KOH, 90%, CAS 1310-58-3, Sigma-Aldrich). Preparation of ¹H NMR samples for liquid products analysis: deuterium oxide (D₂O, 99.9%, CAS 7789-20-0, Sigma-Aldrich), dimethyl sulfoxide (99.9%, CAS 67-68-5, Sigma-Aldrich).

2.2. Synthesis of CuO via hydrothermal route: tabular-prismatic particles

CuO material with prismatic tabular morphology is obtained through hydrothermal route following a modified procedure from the literature [68]. In particular, a copper(II) nitrate solution (2.7 g in 37.5 mL of deionized water) was stirred for 30 min at a temperature of 40 °C. Subsequently, while maintaining the solution in constant agitation, a 6 M NaOH solution is added dropwise to the prepared solution until a pH of around 12 is reached (Vol. approx. 37.5 mL). This way, a black precipitate of CuO is formed at the bottom of the reaction vessel.

The obtained solution (Total Vol. approx. 75 mL) is transferred into a 125-mL sealed Teflon-lined autoclave Model 4748 (Parr Instrument Company, Illinois, USA). The autoclave is hermetically sealed in order to prevent liquid spillage, and thermally treated in a Carbolite CWF1200 muffle furnace (Carbolite, Hope, UK) with the following thermal program: (i) heating ramp from RT to 200 °C (heating rate: 10 °C min⁻¹), (ii) isothermal step at 200 °C for 10 h, and (iii) cooling step from 200 °C to 50 °C (cooling rate: 5 °C min⁻¹).

Once the reaction is completed, the autoclave is removed from the furnace and naturally cooled to RT. The solution is then transferred into a Falcon tube for centrifugation (6000 rpm, 10 min), and washed 3 times with deionized water and once with acetone.

The final precipitate is oven dried under air atmosphere at 80 °C overnight. Lastly, the dried solid is gently crumbled inside an agate mortar, and sample storage inside a closed glass vial.

The obtained sample is defined as H-TP (CuO sample from 6 M NaOH solution, after hydrothermal route), where “H” indicates the hydrothermal route and “TP” refers to the tabular-prismatic shape of the particles.

2.3. Synthesis of CuO via precipitation route: spheroidal particles

CuO material with spheroidal morphology is obtained through precipitation route following a modified procedure from the literature [69]. In particular, a copper(II) nitrate solution (2.4 g in 50 mL of deionized water) was prepared in a reaction flask under constant agitation for about 2 min at RT. Subsequently, a 0.25 M NaOH solution is added dropwise (rate 5 mL min⁻¹) to the prepared solution (vol. 50 mL). The solution is heated to 80 °C and maintained under constant stirring for 0.5 h. Once the reaction is completed, the solution is allowed to naturally cool down to RT. The solution is then transferred into a Falcon tube for centrifugation (6000 rpm, 10 min), and washed 3 times with deionized water and once with acetone.

The final precipitate is oven dried under air atmosphere at 80 °C

overnight. Since in this case it is registered the formation of a bluish-white precipitate, indicating the formation of copper(II) hydroxide-nitrate ($\text{Cu}_2(\text{OH})_3\text{NO}_3$), the solid is transferred into an alumina crucible and thermally treated in a Carbolite CWF1200 muffle furnace (Carbolite, Hope, UK) with the following thermal program: (i) heating ramp from RT to 700 °C (heating rate: 10 °C min⁻¹), (ii) isothermal step at 700 °C for 5 h, and (iii) cooling step from 700 °C to 50 °C (cooling rate: 5 °C min⁻¹). Lastly, the dried solid is gently crumbled inside an agate mortar, and sample storage inside a closed glass vial.

The obtained sample is defined as PT-Sp (CuO sample from precipitation route employing a 0.25 M NaOH solution, after thermal treatment), where “P” indicates the precipitation route, “T” indicates that the sample is thermally treated, and “Sp” refers to the shape of the particles, namely spheroidal.

2.4. Synthesis of CuO via precipitation route: large sheet-like particles

CuO material with sheet-like morphology is obtained through a precipitation route following a modified procedure from the literature [69]. In particular, a copper(II) nitrate solution (2.4 g in 50 mL of deionized water) was prepared in a reaction flask under constant agitation for about 2 min at RT. Subsequently, a 1 M NaOH solution is added dropwise (rate 5 mL min⁻¹) to the prepared solution (vol. 50 mL). The solution is then heated to 80 °C and maintained under constant stirring for 0.5 h. Once the reaction is completed, the solution is allowed to naturally cool down to RT.

The sample obtained employing such NaOH concentration shows the formation of black precipitates indicating the direct formation of CuO. The solution is then transferred into a Falcon tube for centrifugation (6000 rpm, 10 min), and washed 3 times with deionized water and once with acetone.

The final precipitate is oven dried under air atmosphere at 80 °C overnight. Lastly, the dried solid is gently crumbled inside an agate mortar, and sample storage inside a closed glass vial.

The obtained sample is defined as P-Sh (CuO sample from precipitation route employing a 1 M NaOH solution), where “P” indicates the precipitation route, and “Sh” refers to the shape of the particles, namely sheet-like.

2.5. Characterizations

X-ray powder diffraction (XRD) patterns were recorded by means of a Rigaku Miniflex 600 at the University of Milano-Bicocca. The acquisition was performed using a Cu source (40 kV, 15 mA), scanning in the 10–80° 2θ range, with a step size of 0.02°, angular velocity 5.0° per minute. Instrumental PDXL-2 software was used for the sake of comparison with reference diffraction patterns from the ICDD database. The crystallite average size was estimated by means of the Debye-Scherrer equation (Equation (1)):

$$d = \frac{k\lambda}{\beta \cos\theta} \quad (1)$$

where d is the average size of the crystalline domains (expressed in nm), K is the shape factor (typically 0.9), λ is the X-ray wavelength (0.154 nm for a Cu source), β is the line broadening at half the maximum intensity (FWHM) of the selected Bragg angle after subtracting the instrumental line broadening (expressed in radians), and θ is the Bragg angle (expressed in radians). Regarding the CuO systems, the Bragg angle selected was the crystal reflection at $\text{ca. } 2\theta = 35.6^\circ$ corresponding to the Miller index (−111).

Scanning electron microscopy (SEM) micrographs of powder ECs were collected by means of a Zeiss Gemini 500 microscope at the University of Milano-Bicocca equipped with a traditional secondary electrons' detector. Samples were deposited onto SEM stubs using a double-adhesive carbon tape for SEM analysis. Stabs were covered with a gold

coating to prevent any charging effect by means of a VPI SD-900 C sputter coater. Particle size distribution was performed by using ImageJ software, analyzing more than 50 particles per micrograph. SEM images of GDEs have been acquired at the University of Milan by using a FE-SEM Sigma (Zeiss, Germany) operating at 5 kV with a working distance of 5 mm and an Inlens detector. The electrodes were platinum coated by an EM ACE 600 (Leica Microsystem, Germany).

Transmission electron microscopy (TEM) analysis was carried out by means of a FEI TECNAI G2 F20 electron microscope of the Unitech COSPECT at the University of Milan, operating at an accelerating voltage of 200 kV, with a Gatan CCD camera allowing high-resolution imaging. Few mg of the specimens were sonicated in 2-propanol and then transferred as a suspension on a copper grid covered with a holey carbon film. Micrographs were taken after solvent evaporation, spanning over the whole region of the sample, to achieve a truly representative statistical mapping of the investigated materials.

N₂ adsorption-desorption isotherms were recorded at 77 K using a Micromeritics ASAP analyzer. Prior to analysis, the samples were degassed under vacuum at 120 °C. The specific surface area (SSA) was determined applying the Brunauer–Emmett–Teller (BET) model.

Fourier Transform infrared (FT-IR) spectroscopic measurements were performed in transmission mode with a resolution of 2 cm⁻¹ on a Bruker INVENIO FTIR spectrometer equipped with a MCT detector. Samples were prepared as self-supporting pellets with an optical density of $\text{ca. } 13 \text{ mg}\cdot\text{cm}^{-2}$ and placed in a custom-made quartz IR cell with KBr windows. Prior to measurements, each sample was outgassed at 200 °C for 1 h, reaching a residual pressure of $\text{ca. } 5 \cdot 10^{-4}$ mbar. IR spectra of adsorbed CO were then recorded at liquid nitrogen temperature. The spectra were normalized to the optical density of the pellet.

Raman spectra were acquired using a Renishaw inVia Raman Microscope spectrometer, equipped with a 1800 lines mm⁻¹ grating and CCD detector. The excitation source was an Ar⁺ laser emitting at 514 nm. The excitation beam was focused on the sample through a 50 × objective lens. All the spectra were acquired using 10% of laser power (corresponding to 0.5 mW), accumulating 20 spectra of 30 s each.

2.6. Gas diffusion electrode (GDE) preparation

Gas diffusion electrodes (GDE) were prepared by spray coating catalyst ink onto hydrophobic carbon paper (C paper, Freudenberg H23C8). The ink was prepared by suspending 10 mg of ECs powder in 10 mL of isopropanol for 20 min of sonication. Then, a known amount (50 μL) of Nafion™ solution was added, followed by sonication for 10 min at RT. The as-prepared ink was sprayed on a 3 × 3 cm² GDL employing an airbrush gun (Paasche H3AS) with a 0.635 mm nozzle. To guarantee the uniformity and reproducibility of the catalyst layer the GDL was positioned vertically at $\text{ca. } 3 \text{ cm}$ from the nozzle, and the spraying direction was changed frequently, back and forth. To study the effect of the ECs loading on the final performance in CO₂RR, electrodes were prepared for each sample with three different amounts of EC powder: low (1–2 mg), medium (3–4 mg), and high loading (5–6 mg). The exact amount of EC deposited was determined by accurate weighing. Each electrode was thus labelled as X_Y, where: “X” refers to the sample code of the investigated material, whereas “Y” indicates the low (l), medium (m), or high (h) catalyst amount.

2.7. Electrocatalytic measurements

The measurements have been carried out using an electrochemical station (Metrohm Autolab PGSTAT204) connected with a flow cell system. The EC material assembled in a GDE (as described in Section 2.6) was used as the cathode, while an Ag/AgCl electrode (CH Instruments) and Ni foam were employed as reference and counter anode electrode, respectively. In this electrochemical configuration, the parallel disposition of the working and the counter electrode ensures a uniform potential distribution across the cell. The anode and cathode electrodes are

2 cm apart, resulting in a high ohmic resistance that prevents meaningful measurement of the full-cell potential in this configuration. The catholyte and anolyte compartments were filled with a 1 M KOH solution, which was recirculated continuously by a peristaltic pump. CO₂ gas was fed from the backside (the side without the catalytic material) of the gas-diffusion electrode at a flow rate of 50 sccm. An anion-exchange membrane, PET-reinforced (Quintech), was used to separate the anodic and cathodic compartments to avoid oxidation of CO₂RR products at the anode.

Gaseous products were sampled with a gastight syringe and analyzed with an Agilent 8860 gas chromatograph equipped with a thermal conductivity detector (TCD), a flame ionization detector (FID), and two columns, one Coarboxen (to separate permanent gases) and a molecular sieve (MS, to separate C₁–C₄ products) columns. The GC was calibrated for H₂, CO, CH₄, and C₂H₄. Two standard gas mixtures (Sapio) were used for calibration. Ultra-pure Ar (grade 5) was employed as carrier gas. The TCD was used to detect hydrogen (H₂), oxygen (O₂), nitrogen (N₂), and carbon monoxide (CO), while the FID detector was employed to determine methane (CH₄), ethylene (C₂H₄), and, eventually, carbon dioxide (CO₂) through a methanizer.

The Faradaic Efficiency (FE) for the gaseous product is calculated with the following equation (Equation (2)):

$$FE = \frac{n_e \times F \times C \times f}{V_M \times I} \quad (2)$$

where n_e is the number of electrons required to reduce CO₂ to each product, F is the Faraday constant, C is the product concentration in the gas stream obtained from GC analysis (in ppm), f is the flow rate (sccm), V_M is the normal molar volume (24.1 L/mol) and I is the absolute value of the imposed current densities (mA cm^{−2}), ranging from 50 up to 400 mA cm^{−2} and representing the reaction rate in the electrocatalytic systems. In particular, the system has been kept at each chosen current density for at least 5 min, allowing the obtainment of an almost constant working electrode potential, which has been continuously monitored, up to a total measurement time of ca. 25 min, as represented in Figure S1. The potential vs. Ag/AgCl values were converted into the RHE scale using the following equation: $E_{RHE} = E_{AgCl} + 0.059 \text{ pH} + E_{AgCl}^\circ$, with E_{AgCl}° (3 M KCl) = 0.210 V at 25 °C.

Each FE value has been determined by averaging the results obtained by performing two consecutive GC analysis at each fixed applied current density.

Stability properties of selected GDE electrodes have been preliminary checked by monitoring the e-CO₂RR at the highest reaction rate (400 mA cm^{−2}) up to a maximum time of 3 h.

The FE for liquid products has been determined for selected samples by quantitative ¹H NMR analysis. In particular, a volume of 0.2 mL of catholyte, anolyte and cold aqueous trap (collecting species possibly stripped by the CO₂ flow) was sampled after a specific reaction time and mixed with 0.1 mL of a 500 ppm DMSO solution (as the internal standard) and 0.4 mL of D₂O. The mixture was analyzed with ¹H NMR. The equation used for the calculation of the liquid products FE is (Equation (3)):

$$FE = \frac{n_e \times F \times C}{V \times I \times t} \quad (3)$$

where n_e is the number of electrons required to reduce CO₂ to each product, F is the Faraday constant, C is the product concentration obtained from ¹H NMR (in ppm), V is the total volume of the electrolyte, I is the current (mA) and t is the duration of electrolysis (usually 2000 s at 200 mA cm^{−2}). The liquid products evaluation has been performed twice and averaged. The uncertainty associated to the calculated mean FE values resulted to be lower than 5%.

3. Results and discussion

3.1. Structural, morphological and physicochemical characterization of the CuO ECs

Fig. 1 reports the XRD performed on the investigated materials. In general, in all cases (i.e., H-PT, PT-Sp, and P-Sh) the presence of diffraction peaks at ca. $2\theta = 32.5^\circ$ (110), 35.6° (−111), 38.8° (111), 46.3° (−112), 48.7° (−202), 51.4° (112), 53.5° (020), 58.4° (202), 61.6° (−113), 65.9° (022), 66.3° (−311), 68.2° (220), 72.5° (311), and 75.3° (−222) confirms the presence of the monoclinic crystal phase of CuO (ICDD 01–080–1268), in agreement with the literature (Fig. 1) [70]. Interestingly, as reported in Sections 2.2, 2.3, and 2.4, both H-PT and P-Sh samples are directly synthesized in their CuO crystal form, whereas PT-Sp sample, instead, initially precipitates as Cu₂(OH)₃NO₃. The XRD analysis registered for this intermediate form reveals the presence of diffraction peaks at ca. $2\theta = 12.8^\circ$ (001), 21.6° (110), 25.8° (002), 32.0° (200), 33.6° (−201), 36.5° (121), 39.8° (−202), 41.9° (013), 43.5° (122), 49.2° (−131), 51.4° (−123), 53.4° (123), 58.7° (−321), 60.8° (033), and 62.4° (041) attributable to the presence of the crystal phase of Cu₂(OH)₃NO₃ (ICDD 01–075–1779), and in agreement with literature (Figure S2b, SI) [71]. After performing the thermal treatment under air atmosphere, this intermediate form evolves toward the formation of the CuO crystal form, as demonstrated by the XRD analysis (Fig. 1).

The analysis of the cell parameters (a , b , c), and the average crystallite size (d) calculated by employing the Debye-Scherrer equation for all CuO samples is summarized in Table 2.

In general, similar values of cell parameters are registered for all the three samples. Conversely, the calculated average crystallite size varies significantly with the sample preparation. In fact, regarding the precipitation route, the average crystallite size of PT-Sp sample (obtained employing 0.25 M NaOH, but after performing a thermal treatment at high temperature) is remarkably higher compared to the value calculated for P-Sh sample (obtained employing 1 M NaOH, but without performing a thermal treatment at high temperature), showing ca. 50 nm, and ca. 23 nm size, respectively. This phenomenon might be attributable to the effect played by the higher annealing temperature, favoring the grain growth of the final material [72].

Lastly, H-TP sample (obtained by means of hydrothermal route at 200 °C employing 6 M NaOH) shows the highest average crystal size of ca. 70 nm. Since hydrothermal routes are based on crystallization reactions happening into sealed autoclaves at temperatures in the

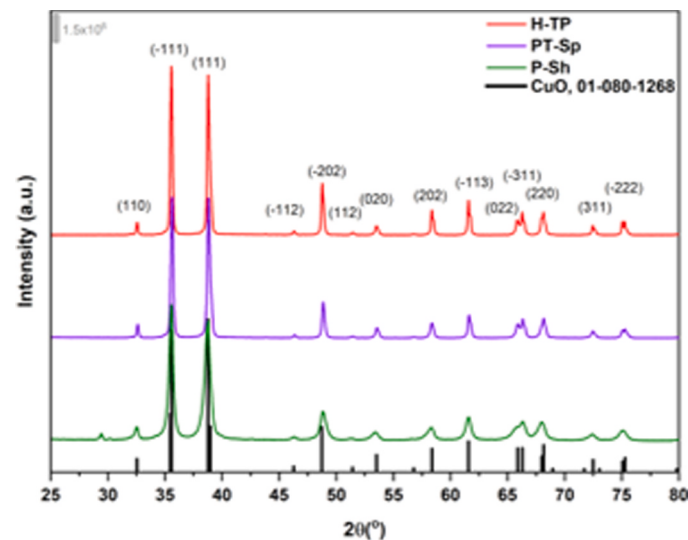


Fig. 1. XRD patterns of the investigated ECs, namely: H-TP sample (red, top), PT-Sp sample (violet, middle), and P-Sh sample (green, bottom). XRD reference pattern: tenorite (CuO, 01–080–1268, black).

Table 2

Average crystallite size and lattice parameters determined by XRD analysis, and calculated BET SSA values for the three CuO ECs.

Samples	d (nm)	a (Å)	b (Å)	c (Å)	BET SSA (m ² g ⁻¹)
H-TP	70	4.68	3.42	5.13	< 1
PT-Sp	50	4.68	3.42	5.12	< 1
P-Sh	23	4.69	3.43	5.14	15

150–300 °C range and relatively high pressure, it is expected a crystallinity increase of the resultant material, in accordance with the literature [73].

Fig. 2 reports SEM and TEM morphological characterizations of the three CuO investigated materials, whereas particle size distribution (calculated by analyzing the SEM micrographs) is reported in Figure S2. As shown in Figs. 2a and 2b, H-TP sample is composed by crystal aggregates with a tabular-prismatic shape and length in the 0.5–1.5 µm range according to both TEM and SEM analysis (Figure S3, SI). Differently, the PT-Sp sample precursor was characterized by the formation of irregularly shaped particle aggregates with mean size in the 60–250 nm range (Figure S2a, SI). Interestingly, upon the application of the reported thermal treatment, PT-Sp sample resulted to be composed by irregular spheroidal particles, with dimensions dispersed in the relatively wide

0.2–1.0 µm range (Figs. 2c, and 2d) according to both TEM and SEM analysis (Figure S3, SI). Lastly, in the case of P-Sh samples, SEM analysis clearly reveals the formation of two-dimensional particle aggregates, exhibiting a peculiar sheet-like morphology, with an average thickness of ca. 40 nm and a mean 0.8–1.5 µm length size (Figs. 2e, and 2f) according to both TEM and SEM analysis (Figure S3, SI).

In agreement with the microscopy data and with the average crystallite sizes obtained from XRD analysis, P-Sh sample exhibits a reasonably high BET specific surface area, whereas it is much more limited for PT-Sp and H-TP samples (Table 2). Therefore, the morphological characterization evidenced that the three selected samples are characterized by the same CuO bulk crystal structure, sub-micrometric particle size (following the order P-Sh > H-TP > PT-Sp, with P-Sh having a nanometric thickness), but significantly different particle aggregates morphology, passing from irregularly spheroidal toward tabular-prismatic or two-dimensional sheet-like shapes for the PT-Sp, H-TP and P-Sh samples, respectively.

3.2. Electrocatalytic CO₂RR testing

Electrochemical results obtained with the investigated GDEs containing variable EC loadings (either low, medium or high) are reported in Fig. 3 in terms of Faradaic Efficiencies (FEs) toward gaseous CO₂

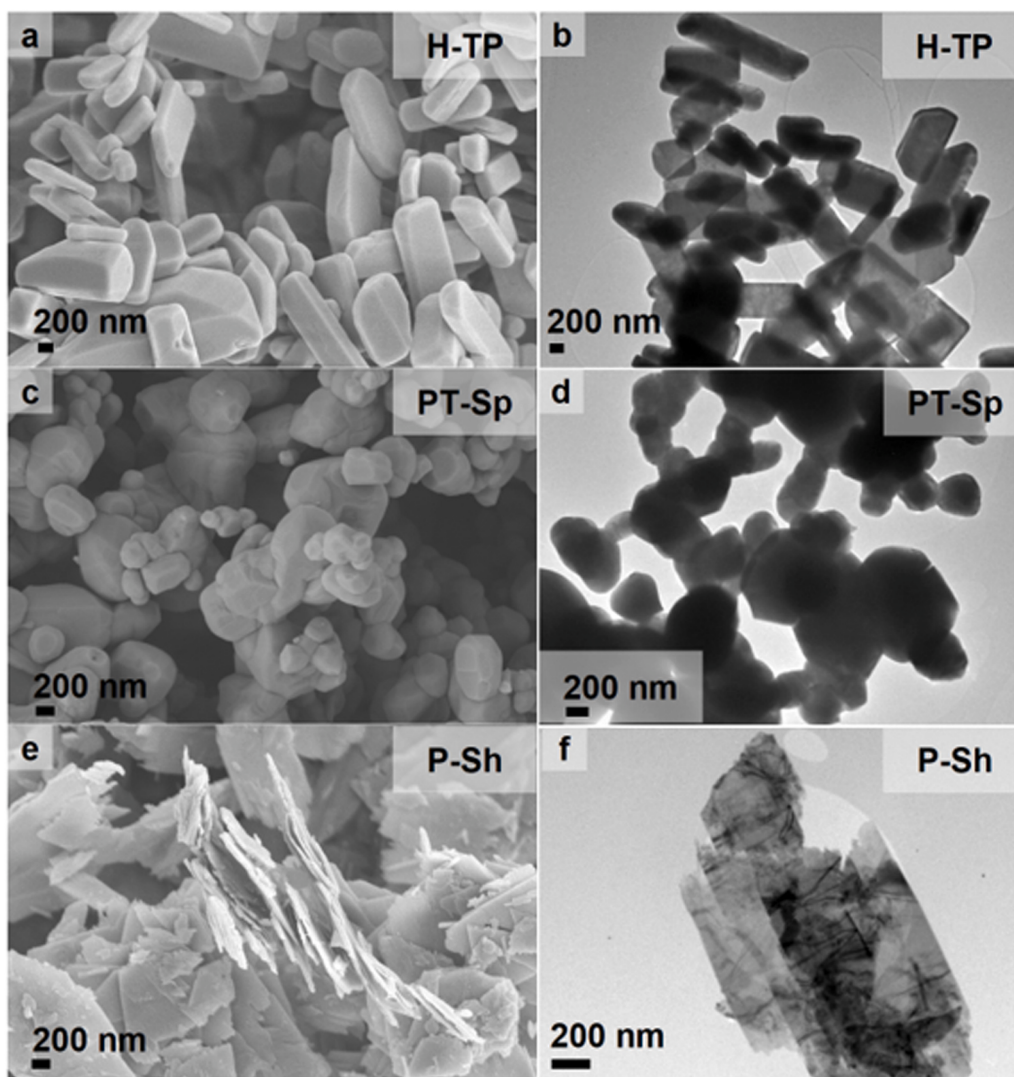


Fig. 2. SEM (left) and TEM (right) micrographs of the investigated ECs, namely: Panels (a, b) H-TP sample (hydrothermal route; top), (c, d) PT-Sp sample (precipitation route with 0.25 M NaOH and thermal treatment; middle), and (e, f) P-Sh sample (precipitation route with 1 M NaOH; bottom).

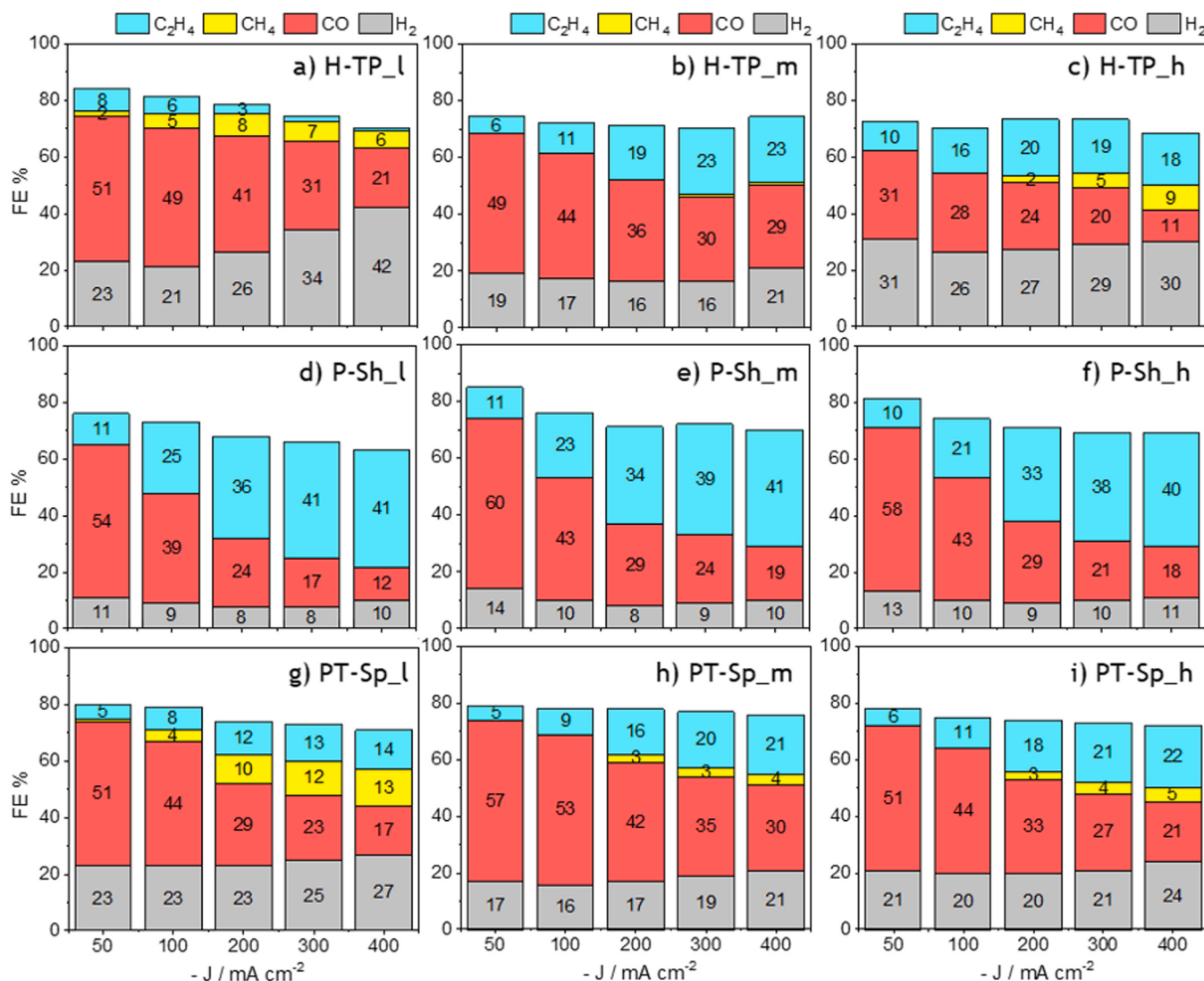


Fig. 3. Faradaic Efficiencies (FE) toward gas products (ethylene, carbon monoxide, methane, and hydrogen) during the electrocatalytic CO_2 reduction at different current densities (100, 200, 300, and 400 $mA\ cm^{-2}$) obtained with (a-c) H-TP, (d-f) P-Sh and (g-i) PT-Sp samples at low (left data column), medium (middle data column) and high (right data column) loading.

reduction products.

The FEs obtained with H-TP sample, synthesized via the hydrothermal method, confirm the good efficiency of CuO NPs in catalyzing the electrochemical CO_2 RR, mainly toward CO production, in line with what already established in the literature [74–76], though the undesired H_2 production resulted to be quite significant, especially at relatively low catalyst loading (1–2 mg), where the FE toward H_2 reaches values of ca. 30%, probably due to a low catalyst surface exposure of active sites available for promoting the CO_2 RR. At the same time, the progressive increase of ECs loading significantly promotes the ethylene formation (almost independently on the applied current density), although the total FE for CO_2 reduction tends to diminish, probably due to a relatively higher electrical and CO_2 diffusion resistance attained with the thicker catalyst layer (CL). In fact, both SEM and TEM images evidenced that H-TP sample is characterized by large and densely packed particles. It is well known that both geometry and morphological features of the CL can affect CO_2 transport properties within the GDE and, thus, the overall performance in terms of FEs for CO_2 RR [77,78].

Optimal results have been attained with the GDE prepared by employing a medium (3–4 mg) catalyst loading (Fig. 3b), possibly allowing the obtention of a proper metal oxide layer thickness guaranteeing an optimized electrical transfer and CO_2 diffusion.

Interestingly, the P-Sh sample, characterized by a sheet-like morphology, exhibits completely different catalytic properties, with an extremely relevant ethylene production (FE of 35–40%) on the whole investigated catalyst loading range and almost independently on the applied 200–400 $mA\ cm^{-2}$ current density range. In particular, the almost exclusive CO production (FE of 60–70%) at low current densities tends to decrease in favor of ethylene production by increasing the applied current density, as expected for the C–C coupling pathway, i.e. CO being considered as the major intermediate for the formation of C_2+ hydrocarbons via CO_2 RR. In fact, according to both experimental research and density functional theory (DFT) calculations, *CO – *CO coupling is the most kinetically favorable pathway for C–C coupling leading to C_2+ reduction products on Cu-based catalysts [79]. For this reason, the accumulation of a high local concentration of CO species on the catalyst surface is necessary to enable the coupling and, consequently, the production of ethylene. This condition is achieved particularly efficiently at high current densities, when the CO_2 RR proceeds more rapidly, thus resulting in a higher local production of activated intermediates.

P-Sh sample ensures a total conversion towards gaseous products above 70% with a very high selectivity corresponding to the quite exclusive production of ethylene, with a FE above 40% at 400 $mA\ cm^{-2}$

in 1 M KOH electrolyte. Importantly, the beneficial effect intrinsically played by the CuO sheet-like morphology in favoring ethylene production, also allowing stable catalytic activity, is demonstrated by the fact that P-Sh sample guarantees similar and satisfying performances across all the three investigated catalyst loadings (1–5 mg).

Differently, thermally treated PT-Sp sample shows an intermediate catalytic activity between those shown by H-TP and P-Sh samples. In particular, the total conversion towards gaseous species remains consistently stable at ca. 80%, with CO being the predominant product. As in the previous cases, ethylene production increases with increasing the applied current density; however, the highest attained FE for C₂H₄ corresponds to ca. 20% and is almost half of that exhibited by the sheet-like P-Sh sample under similar experimental conditions (*i.e.*, by applying 400 mA cm⁻² current density with a medium or high catalyst loading). The undesired and competitive hydrogen production results instead to be favored and almost double compared to that attained with the best performing P-Sh catalyst (see Fig. 3h vs. Fig. 3e).

Therefore, PT-Sp sample shows broader products distribution and lower ability toward CO₂ (instead of water) electroreduction on the whole investigated catalyst loading range. This finding may be related to the CuO particles morphological changes from sheet-like structures to irregular spherical shapes observed upon the applied heating procedure.

In this scenario, De Gregorio *et al.* [80] found a crucial role played by Cu nanoparticles shape among spherical, octahedral, and cubic morphology in affecting the overall CO₂RR activity. In particular, 44 nm-length Cu cubes achieved the highest selectivity toward CO₂RR, with a total FE of 80% and selectivity for ethylene of ca. 41% in a flow-cell electrolyzer and the authors suggested the optimal ratio between edge and (100) plane sites (of ca. 0.25), attained with such material, as a crucial structural parameter for controlling the CO₂RR product selectivity [80].

Interestingly, in our investigation only PT-Sp sample exhibits a detectable production of methane, with a maximum of FE of ca. 13% at 400 mA cm⁻² with a low catalyst loading (Fig. 3g). This observation suggests a possible effect induced by the electrode polarization. It has been reported that at high overpotentials (more negative than -0.8 V vs. RHE), both ethylene (C₂H₄) and methane (CH₄) can form simultaneously from the common COH intermediate on Cu (100) and Cu (111) facets [63,81], though this behavior has not been generally observed in alkaline media [81–83]. We thus hypothesize that the relatively more negative potential required to achieve the target current density at low (1–2 mg) catalyst loading, where the exposed catalyst surface is limited, may favor methane over ethylene formation. This change of selectivity may be attributed to modifications in the local reaction environment under high polarization conditions, also including variations in local pH, CO₂ availability, and surface coverage of reaction intermediates.

In this scenario, the better catalytic properties attained by the P-Sh_m sample is also demonstrated by the relatively less negative and more stable average potential values attained at each target current density, especially if compared to those exhibited by the H-TP_m electrode,

originally composed by tabular-prismatic CuO particles (see Table S1).

Aiming at further highlighting the catalytic differences among the investigated CuO-based materials, the partial current densities of H₂, CO and C₂H₄ (indicated as J_{H₂}, J_{CO} and J_{C₂H₄}) attained with the corresponding GDEs prepared with a medium catalyst loading can be directly compared in Fig. 4.

Firstly, the undesired hydrogen production, representing a serious parasite reaction because occurring at the expense of the FE for CO₂RR, resulted to proceed differently on the tested samples. In particular, while H-TP and PT-Sp catalysts exhibit a relatively high J_{H₂}, with a sharp increase above the application of a total current density of 200 mA cm⁻², the most efficient P-Sh clearly minimizes hydrogen evolution, by keeping J_{H₂} significantly below 40 mA cm⁻² on the whole investigated total applied current density (see Fig. 4a).

Concerning the catalytic evolution of the two main CO₂-derived gaseous products, while J_{CO} tends to progressively slow down (Fig. 4b), J_{C₂H₄} almost linearly increases by applying a more negative total current density (Fig. 4c).

The observation that operating at more cathodic potential favors C–C coupling over C₁ products pathway agrees with literature and explains the systematic increase of FE_{C₂H₄}/FE_{CO} ratio attained with all the catalysts as the potential becomes more negative. Moreover, the complete absence of any partial current density for ethylene production upon the application of a total 50 mA cm⁻² (unlike to what observed for the other gaseous products) demonstrates that the C–C coupling mechanism requires a minimum activation current. This threshold is necessary to drive the CO₂ reduction at a rate high enough to generate a proper accumulation of activated CO intermediates.

Further confirmation of the ability exhibited by the investigated materials in favoring the CO₂ conversion into C₂⁺ species is obtained from liquid products analysis via ¹H NMR analysis, especially for the P-Sh sample. In fact, as reported in Table 3, a FE for ethanol of ca. 12%, together with traces of propanol (6%) and acetate (1%), has been attained in the electrochemical CO₂RR performed at 200 mA cm⁻² with P-Sh_m sample. Lower overall FE towards liquid and gaseous C₂⁺ species have been attained with both PT-Sp_m and H-TP_m samples.

Table 3

Faradaic Efficiencies for liquid products (formic acid, acetic acid, ethanol and propanol) collected during the CO₂ electrocatalytic reduction at 200 mA cm⁻² with P-Sh_m, PT-Sp_m and H-TP_m samples. The uncertainty associated to the reported mean FE values resulted to be lower than 5%.

Liquid products	FE at 200 mA cm ⁻² (%)		
	P-Sh _m	PT-Sp _m	H-TP _m
Formic acid	4	5	4
Acetic acid	1	1	1
Ethanol	12	11	10
Propanol	6	4	2
Total	23	22	17

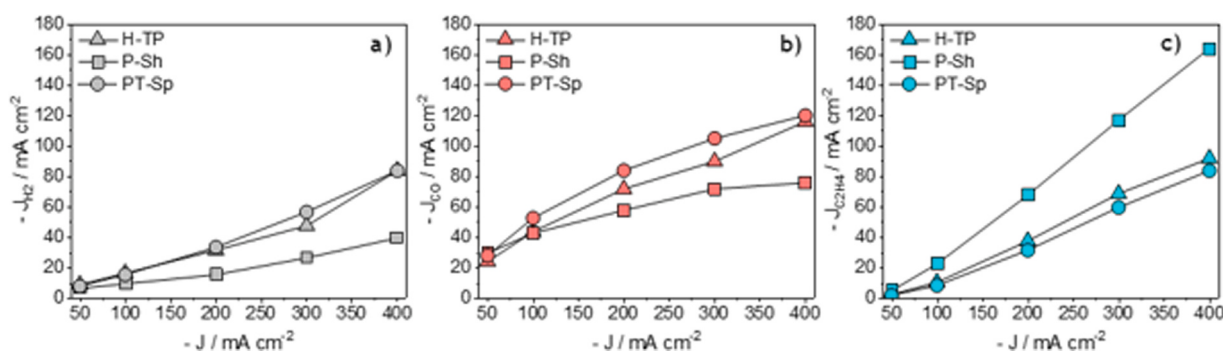


Fig. 4. Partial current density for a) H₂, b) CO and c) C₂H₄ production attained with H-TP_m (triangle symbols), P-Sh_m (square symbols) and PT-Sp_m (circle symbols) electrodes.

Importantly, this kind of analysis allowed to close the total FE balance to ca. 100%, verifying the effective production of other C_{2+} liquid hydrocarbons, thus excluding any significant underestimation of H_2 , which may not be detected if it accumulates into the catholyte solution without being released in the gas phase compartment of the here adopted flow cell.

Furthermore, preliminary tests confirmed the excellent ability exclusively exhibited by the best performing P-Sh_m electrode in favoring the CO_2 reduction with a satisfying stability, guaranteeing the obtainment of almost unvaried FE towards C_2H_4 product (ca. 40–45%) up to 3 h of reaction at 400 mA cm^{-2} (see Figure S4 and Table S2). Differently, the catalytic performance of both PT-Sp_m and H-TP_m materials started to get worst relatively early, as demonstrated by the noisy potential signal registered few minutes after reaching the target 400 mA cm^{-2} current density (see Figure S4). This behavior may be ascribed to the structural reconstruction experienced by such materials, undergoing to severe aggregation (*vide* Section 3.4), leaving uncovered area of the underlying C paper, which may undermine the overall GDE stability, thus inducing the undesired flooding, that is the electrolyte permeation across the C paper, accompanied by H_2 production, as supported by the FE data attained during their stability tests and collected in Table S2.

3.3. FT-IR spectroscopy of adsorbed CO

Fourier Transform Infrared (FT-IR) spectroscopy using probe molecules was employed to perform a detailed surface characterization of the samples. The selected probe molecule was carbon monoxide (CO) since it is sensitive to the oxidation state and local environment of the Cu surface sites [84,85]. Before the spectroscopic measurements, the ECs were outgassed at $200\text{ }^\circ\text{C}$ to remove the physisorbed water, thus allowing the adsorption of the CO probe molecules.

As shown in Fig. 5, the spectra of P-Sh are dominated by an intense band centered at 2110 cm^{-1} , accompanied by a higher-frequency shoulder at 2135 cm^{-1} and by a very weak component in the $2200\text{--}2190\text{ cm}^{-1}$ region, detectable mainly at the highest CO equilibrium pressure. For PT-Sp, by contrast, a much broader and less intense band envelope is observed, with a maximum at 2095 cm^{-1} and no clearly resolved high-frequency contributions. Finally, the extremely low specific surface area of the H-TP sample prevented the detection of signals from adsorbed CO.

The band positions suggest that the two materials expose different

copper surface environments and comparison with single-crystal copper literature provides a useful frame of reference. On metallic Cu single crystal surfaces, the singleton C–O stretching frequency of linearly adsorbed CO falls in the range $2085\text{--}2075\text{ cm}^{-1}$ for well-coordinated terrace sites on the most stable Cu (111) and (100) surfaces [86]. The $\nu(CO)$ frequency is sensitive to the local coordination number of the adsorbing Cu atom: for instance, for the Cu (211) surface the signal is blueshifted to 2093 cm^{-1} for step sites and even to 2110 cm^{-1} for kink sites [87]. Also a partial or full oxidation of the Cu surface produces a blueshift of the signals which appear in the $2140\text{--}2110\text{ cm}^{-1}$ range [87]. Based on these literature results, the broad absorption centered at 2095 cm^{-1} in PT-Sp can be thus assigned to adsorption on metallic or partially reduced low-coordination Cu sites. While the main component observed for P-Sh at 2110 cm^{-1} lies at the upper edge of the range usually associated with metallic Cu and therefore is more safely interpreted as arising from cationic sites, like $Cu^{\delta+}$ or Cu^+ [88,89].

The higher-frequency components observed for P-Sh further support the presence of cationic copper species. Literature on CO adsorption over Cu-containing materials and CuO-supported systems assigns bands in the ca. 2130 cm^{-1} range to Cu^+-CO monocarbonyls [88], whereas weak features between 2210 and 2180 cm^{-1} are characteristic of unstable $Cu^{2+}\text{--}CO$ adducts, usually detectable only at low temperature and rapidly removed upon CO evacuation [90].

The spectroscopic results suggest that the investigated CuO morphologies do not simply differ in geometric shape, but also in the nature and distribution of accessible Cu surface sites. In particular, the larger contribution of cationic Cu species observed for P-Sh indicates a surface environment substantially different from that of PT-Sp. This observation, combined with the superior C_2H_4 selectivity exhibited by P-Sh, supports the hypothesis that the EC morphology influences the catalytic response through both geometric and electronic effects. Therefore, the enhanced e- CO_2 RR performance of the sheet-like EC (*i.e.*, P-Sh) seems the result of a combined morphology–surface chemistry relationship rather than as a merely shape-only effect.

3.4. GDE characterization

After assembling the GDE via spray coating, their morphology was studied with SEM analysis at different magnifications to assess whether the shape of the starting powders was retained after the spray and to verify that an even, complete coverage of the C paper was achieved. To do so, electrodes with the highest loading were imaged for all

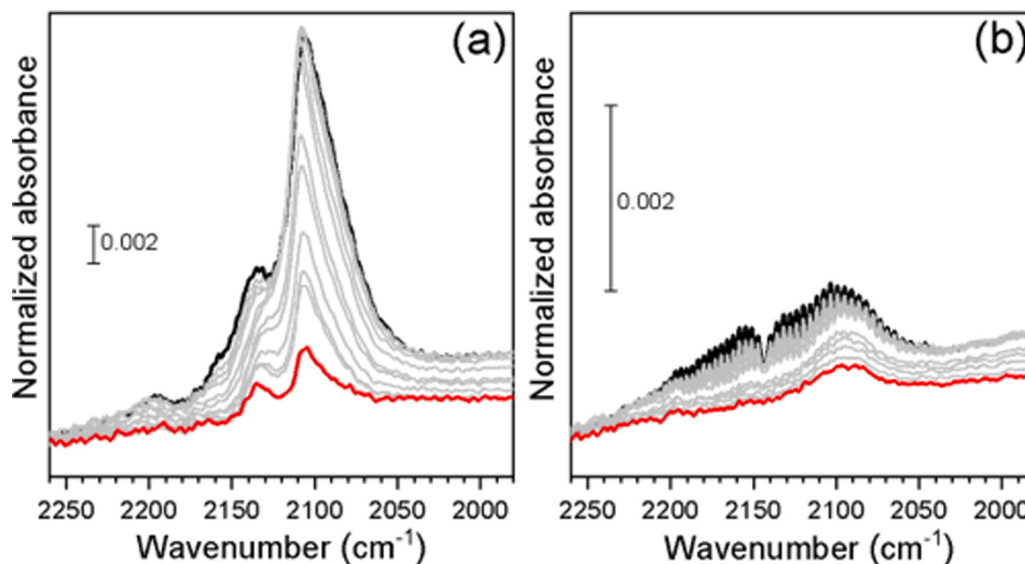


Fig. 5. FT-IR spectra of CO adsorbed at ca. 100 K at different coverages starting from 3 mbar (black curves) with stepwise outgassing till $5 \cdot 10^{-4}$ mbar (red curves) for samples a) P-Sh and b) PT-Sp. The spectrum of the sample before dosing CO was subtracted from all the spectra.

investigated samples (Fig. 6).

The shape of the H-TP sample is observable in Fig. 6a–c. Moreover, this loading was suitable to achieve satisfactory coverage of the underlying C paper, with only a few spots leaving it exposed. The highest-magnification image shows that Nafion wraps the nanoparticles with a continuous coating, which smooths the sharp edges observed in the SEM images of the pure nanoparticles (Fig. 2a).

On the other hand, the same loading of PT-Sp nanoparticles is unable to fully coat the C paper, which appears largely uncoated (Fig. 6d–e), with its pattern observable at the highest magnification (Fig. 6f). Therefore, the sheet-like nanoparticles cover the C paper more effectively due to their in-plane assembly, with the particles facing each other's largest crystal plane, thus resulting in a homogeneous and continuous layer. Conversely, the spherical particles retain the same tendency to agglomerate as observed in the original powder sample (Fig. 2c–d), thereby forming large clusters and leading to a heterogeneous coverage.

Leveraging the ability of the plate-like nanoparticles to stack on their largest facet, the P-Sh sample, due to its 2D sheet-like architecture, enables the complete coating of the C paper. Indeed, SEM images of the P-Sh GDE lack the typical features of the C paper (Figure S5, SI) at all the investigated magnifications (Fig. 6g–i).

SEM images of a GDE prepared with optimized medium P-Sh loading (Fig. 7a, b) show a full coverage of the underlying C paper, attesting to the excellent ability of this sample architecture to spread effectively thanks to its 2D shape.

Ex-post SEM images of the same P-Sh_m GDE indicate that the pristine catalyst undergoes severe reconstruction during the e-CO₂RR, as a consequence of the reduction of copper oxide under reaction conditions [91–93]. Indeed, after reaction, we can clearly appreciate that the initial

micrometric-length sheets are composed of small, sphere-like nanoparticles with mean size of few nm, as predicted by XRD analysis (see Table 2), though P-Sh sample retain the original sheet-like structure network (Fig. 7c). The presence of a continuous nanoparticles network and the evidence that almost all sheets underwent this considerable change testify that all nanoparticles are electrically conductive during the CO₂ reduction and can participate in the reaction (Figures S6, SI). Interestingly, even after prolonged e-CO₂RR, the P-Sh_m electrode retained its sheet-like structural integrity (Figure S7, SI), thus guaranteeing stable catalytic performance, as previously detailed.

Differently, both PT-Sp_m and H-TP_m underwent significant modifications of their original spheroidal or tabular-prismatic particle shapes. In fact, ex-post SEM images of PT-Sp_m and H-TP_m GDEs clearly show that the applied electrochemical reductive conditions dramatically damage both their particles original structural integrity and distribution on the C paper (Figures S8–S9, SI). In both cases, the EC particles tend to coalesce up to the formation of a structureless layer of CuO-based material surrounded by uncovered areas of C paper, negatively affecting the overall e-CO₂RR efficiency.

For a more comprehensive characterization of the post-reaction electrodes, Raman spectroscopy measurements were performed. As shown in Fig. 8a, the Raman spectrum of the electrode before reaction clearly displays bands at 299, 346, and 631 cm^{−1}, which can be assigned to CuO vibrational modes, with symmetry A_g, B_g and B_g, respectively [94]. In contrast, in the spectrum of the post-reaction sample (Fig. 8b), different bands are observed at 150, 218, 415 and 640 cm^{−1}, indicating the formation of Cu₂O. It should be noted that, for ideal cubic cuprite Cu₂O, only the T_{2g} mode is first-order Raman allowed, whereas several commonly observed bands arise from defect-induced activation of IR-active or silent modes, as well as from second-order Raman

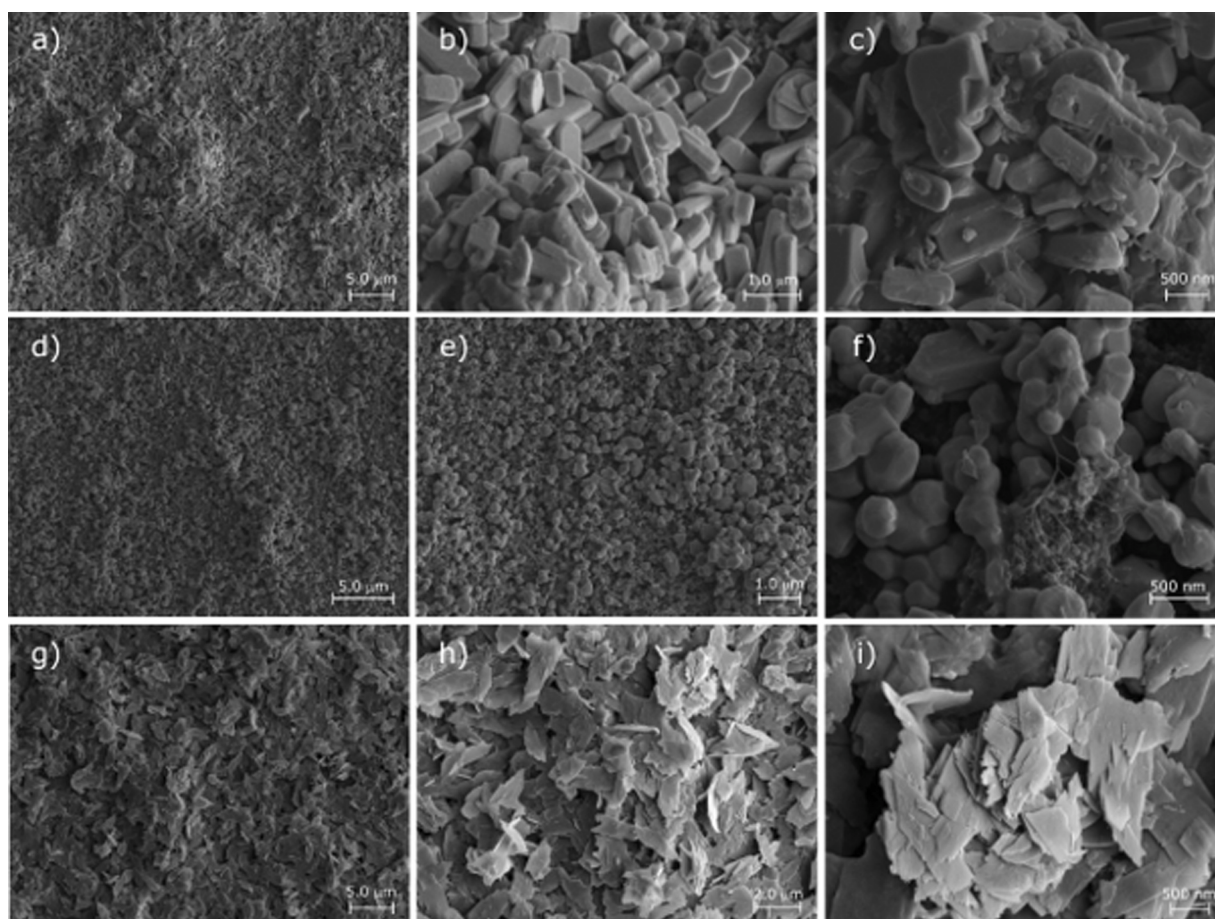


Fig. 6. SEM images of the (a–c) H-TP_h, (d–f) PT-Sp_h and (g–i) P-Sh_h GDEs at different magnifications with the corresponding scale bars.

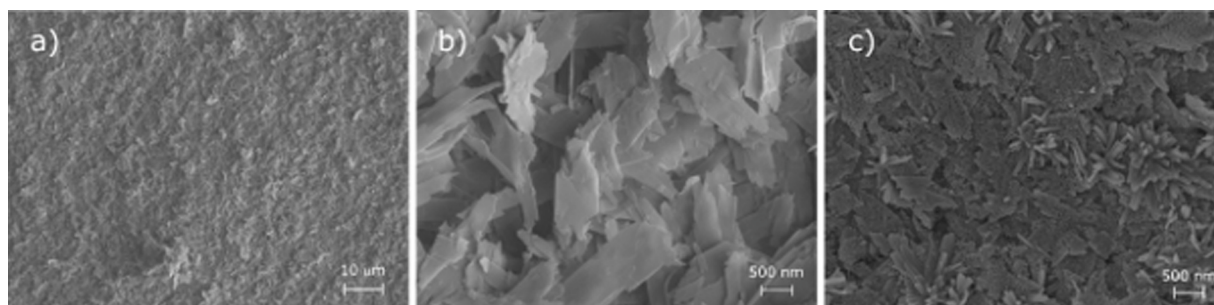


Fig. 7. SEM images of the P-Sh_m GDE (a-b) before and c) after performing the e-CO₂RR. The scale bars refer to adopted magnifications.

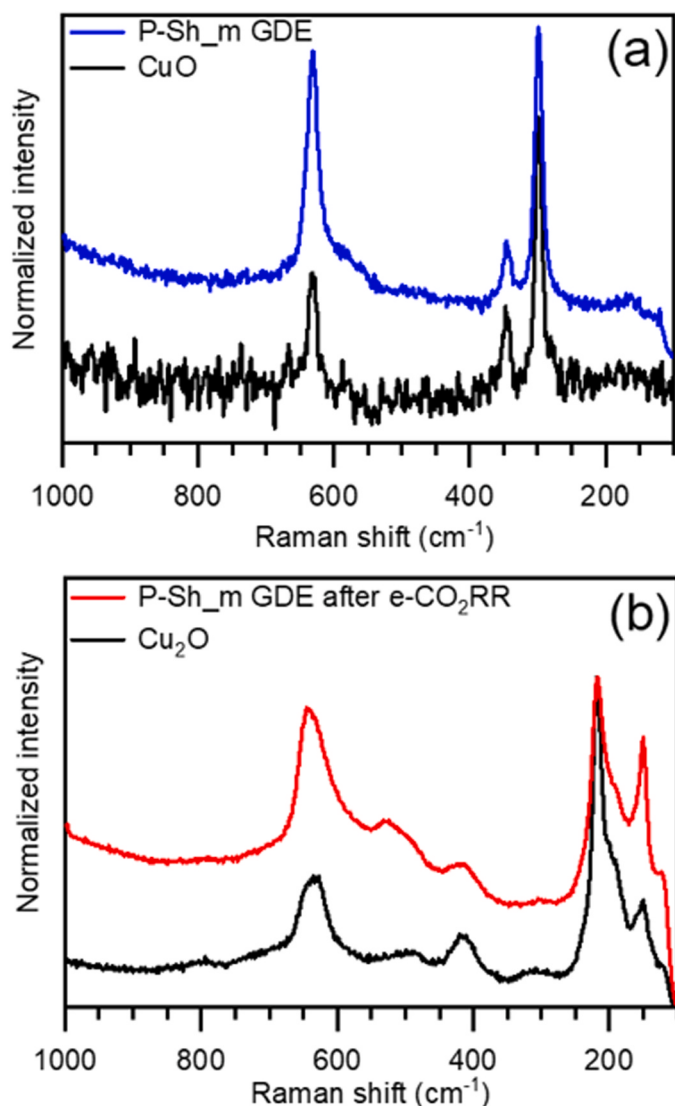


Fig. 8. Raman spectra of the P-Sh_m GDE (a) before and (b) after performing the e-CO₂RR, compared to commercial pure CuO and Cu₂O samples.

scattering. Accordingly, the band at 150 cm⁻¹ can be attributed to a defect-activated T_{1u} mode, while the intense feature at around 218 cm⁻¹ is commonly assigned to a second-order Raman mode of Cu₂O [95]. The bands at 415 cm⁻¹ can be tentatively ascribed to a defect-activated Cu₂O mode, whereas the feature at 640 cm⁻¹ is associated with high-frequency T_{1u} mode [96]. The appearance of these Cu₂O-related bands after e-CO₂RR indicates the formation of reduced copper oxide phases upon electrocatalysis. This modification in the

Raman spectrum after reaction is fully consistent with the modifications observed by SEM and in agreement with previous studies reporting the formation of reduced Cu phases during electrocatalytic CO₂ reduction [91–93].

Taken together, these findings support that the high activity of P-Sh sample stems from its ability to form a continuous coating on the C paper, which provides abundant and proximal catalytically active sites, promoting the C–C coupling, resulting in a 52% FE for total C₂₊ products (gas and liquid) at 200 mA cm⁻², largely overperforming the other two samples (32% FE total C₂₊ products). However, the improved performance of the P-Sh EC cannot be solely attributed to its 2D morphology. In fact, morphology influences several EC properties simultaneously, including SSA, accessible Cu surface sites, EC layer homogeneity, and local mass-transport conditions. The complete coverage of the GDE achieved by the sheet-like system generates a highly interconnected catalytic network that should facilitate the accumulation of CO-derived intermediates and increase the probability of occurrence of C–C coupling reactions. Conversely, spheroidal and tabular-prismatic systems form less homogeneous EC layers and expose a lower density of accessible Cu active sites, resulting in a minor CO₂RR C₂-selectivity and an enhanced HER. These observations indicate that the initial morphology affects the catalytic performance through its impact on the EC architecture and local reaction environment rather than through a purely geometric effect.

4. Conclusions

Three CuO materials with different morphology were synthesized and tested as electrocatalysts for CO₂RR. Despite sharing the same bulk CuO crystal phase, the samples exhibited marked differences in crystallite size, specific surface area, exposed surface sites, and catalyst-layer organization on the GDE, which strongly affected their catalytic behavior.

The sheet-like P-Sh sample displayed the highest specific surface area and the best overall performance, combining high selectivity toward C₂ products and effective suppression of HER. FT-IR spectroscopy of adsorbed CO indicate that P-Sh exposes a larger amount of spectroscopically distinct adsorption sites, including Cu⁺ and a minor fraction of Cu²⁺ centers, whereas PT-Sp is characterized by a heterogeneous distribution of more metallic Cu environments. Moreover, SEM analysis of the GDEs revealed that the 2D sheet-like architecture of P-Sh enables a more homogeneous catalyst coating, likely leading to a higher density of proximal active sites, which favors the formation of multi-carbon products.

Overall, this work demonstrates that the electrocatalytic behavior of oxide-derived Cu ECs is governed by a complex interplay between morphology, surface chemistry, and EC layer architecture. Although the sheet-like P-Sh system exhibited the highest selectivity toward C₂ products, the present study reveals that this behavior cannot be explained solely by its 2D morphology. Instead, the improved performance originates from the simultaneous presence of a higher accessible surface area, distinct Cu surface environments identified by FT-IR

spectroscopy, and the formation of a homogeneous EC coating on the gas diffusion electrode that facilitates the generation and coupling of reaction intermediates.

Furthermore, this study also demonstrated that in the case of sheet-like system, the initial CuO morphology is retained even after prolonged e-CO₂RR operating conditions, generating a highly interconnected catalytic network that promotes C–C coupling reactions enhancing the formation of C₂ products. Vice versa, both tabular-prismatic and spherical systems gave an important structural reconstruction, thus leading to a less favorable EC layer organization and lower e-CO₂RR selectivity.

These findings provide a comprehensive correlation on the structure–property performance of oxide-derived Cu ECs and offer practical guidelines for the rational design of Cu-based ECs employable in e-CO₂RR processes.

CRediT authorship contribution statement

Rodriguez-Flores Tatiana: Writing – review & editing, Investigation, Formal analysis. **Davide Melotto:** Writing – review & editing, Investigation, Formal analysis. **Laura Vigni:** Investigation, Formal analysis, Data curation. **Falak Shafiq:** Writing – review & editing, Writing – original draft, Investigation, Formal analysis. **Roberto Nisticò:** Writing – review & editing, Writing – original draft, Supervision, Resources, Investigation, Formal analysis, Conceptualization. **Maria Vittoria Dozzi:** Writing – review & editing, Writing – original draft, Supervision, Resources, Investigation, Formal analysis, Conceptualization. **Lorenzo Mino:** Writing – review & editing, Writing – original draft, Supervision, Resources, Investigation, Formal analysis, Conceptualization. **Ivan Grigioni:** Writing – review & editing, Writing – original draft, Supervision, Investigation. **Elizaveta Kozyr:** Writing – review & editing, Investigation, Formal analysis.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at [doi:10.1016/j.mtcata.2026.100147](https://doi.org/10.1016/j.mtcata.2026.100147).

Data availability

Data will be made available on request.

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